

Structure Factor

Textbook's convention:
$$F_{\mathbf{g}} = \sum_{m \text{ atoms}} f_m \cdot e^{2\pi i \mathbf{g} \cdot \mathbf{d}_m} \quad f_m = f^{(m)}(s)$$

(Note: form factor are tabulated as functions of $s=g/2$)

Atomic coordinates:
$$\mathbf{d}_m = x_m \mathbf{a}_1 + y_m \mathbf{a}_2 + z_m \mathbf{a}_3 = (x_m, y_m, z_m)$$

(expressed as fractions of basis vectors)

RLV:
$$\mathbf{g} = h\mathbf{b}_1 + k\mathbf{b}_2 + l\mathbf{b}_3$$

Notice that:
$$\mathbf{g} \cdot \mathbf{d}_m = hx_m + ky_m + lz_m$$

A general result:
$$F_{hkl} = \sum_{m \text{ atoms}} f_m e^{2\pi i (hx_m + ky_m + lz_m)}$$

Example:
Simple cubic (*sc*) with two-atom basis

The basis vectors are:

$$\mathbf{a}_1 = a\hat{\mathbf{x}}, \mathbf{a}_2 = a\hat{\mathbf{y}}, \mathbf{a}_3 = a\hat{\mathbf{z}}$$

The “basis” sometimes refers to all the atoms in the unit cell.

$$\text{First atom:} \quad f_A \quad \mathbf{d}_1 = (0, 0, 0)$$

$$\text{Second atom:} \quad f_B \quad \mathbf{d}_2 = (x, y, z)$$

$$F_{hkl} = f_A e^{i0} + f_B e^{2\pi i(hx+ky+lz)} = f_A + f_B e^{2\pi i(hx+ky+lz)}$$

This is the structure factor for *any* integers (*hkl*).

Example:

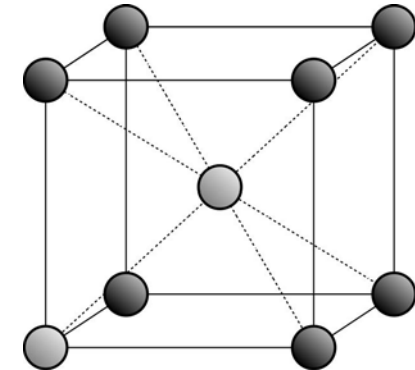
Body-centered cubic (*bcc*) structure

The *bcc* structure can be generated using a *sc* lattice with a two-atom basis.

First atom: f $\mathbf{d}_1 = (0,0,0)$

Second atom: f $\mathbf{d}_2 = (0.5,0.5,0.5)$

Structure Factors: $F_{hkl} = f \cdot \left[1 + e^{\pi i(h+k+l)} \right]$



$$e^{\pi i(h+k+l)} = \begin{cases} 1, & h+k+l = \text{even} \\ -1, & h+k+l = \text{odd} \end{cases}$$

So, $F_{hkl} = \begin{cases} 2f, & h+k+l = \text{even} \\ 0, & h+k+l = \text{odd} \end{cases}$

systematic absences

allowed reflections:

{110}

{200}

{202}

{222}

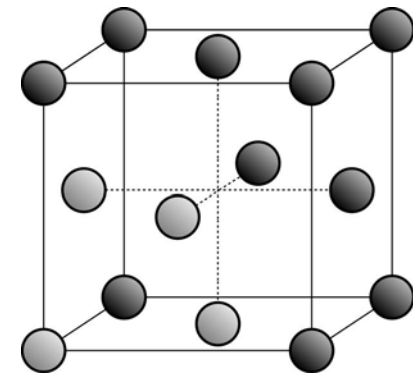
{400}

...

Example: Face-Centered Cubic (fcc) Structure

The fcc structure can be generated from a sc lattice
with a four-atom basis.

First atom:	f	$\mathbf{d}_1 = (0, 0, 0)$
Second atom:	f	$\mathbf{d}_2 = (0, 0.5, 0.5)$
Third atom:	f	$\mathbf{d}_3 = (0.5, 0, 0.5)$
Fourth atom:	f	$\mathbf{d}_4 = (0.5, 0.5, 0)$



Structure Factors: $F_{hkl} = f \cdot \left[1 + e^{\pi i(k+l)} + e^{\pi i(h+l)} + e^{\pi i(h+k)} \right]$

So,
$$F_{hkl} = \begin{cases} 4f, & h, k, l \text{ all even or all odd} \\ 0, & h, k, l \text{ mixed even/odd} \end{cases}$$

allowed reflections:

$\{200\}$
 $\{111\}$
 $\{220\}$
 $\{311\}$
 $\{400\}$

...

Crystal Potential and Structure Factors

Crystal Potential:

$$\Phi(\mathbf{r}) = \sum_{\mathbf{g}} \Phi_{\mathbf{g}} e^{-2\pi i \mathbf{g} \cdot \mathbf{r}}$$

Fourier Components:

$$\Phi_{\mathbf{g}} = \phi(\mathbf{g}) X_{\mathbf{g}}$$

Crystal Function:

$$X_{\mathbf{g}} = 1/v \quad (\text{when } \mathbf{g} \text{ is an RLV})$$

Unit-Cell Potential:

$$\phi(\mathbf{g}) = \sum_{m \text{ atoms}} \phi_m e^{2\pi i \mathbf{g} \cdot \mathbf{d}_m}$$

Atomic Form Factors:

$$f_m = \frac{2\pi m e}{h^2} \phi_m$$

Structure Factors:

$$F_{\mathbf{g}} = \sum_{m \text{ atoms}} f_m e^{2\pi i \mathbf{g} \cdot \mathbf{d}_m}$$

Structure Function:

$$U_{\mathbf{g}} = \frac{2m e}{h^2} \Phi_{\mathbf{g}} = \frac{F_{\mathbf{g}}}{\pi v}$$

The Fourier components of the crystal potential are normalized by the unit-cell volume.

Diamond structure

Two identical, interpenetrating *fcc* lattices offset by $\frac{1}{4}[111]$

First lattice: $\mathbf{d}_1 = \mathbf{0} = 0 \cdot \mathbf{a}_1 + 0 \cdot \mathbf{a}_2 + 0 \cdot \mathbf{a}_3 = (0, 0, 0)$

$$F_{hkl}^{(1)} = f \cdot \left[1 + e^{\pi i(k+l)} + e^{\pi i(h+l)} + e^{\pi i(h+k)} \right]$$

Second lattice: $\mathbf{d}_2 = a(\hat{\mathbf{x}} + \hat{\mathbf{y}} + \hat{\mathbf{z}})/4 = (\mathbf{a}_1 + \mathbf{a}_2 + \mathbf{a}_3)/4 = (1/4, 1/4, 1/4)$

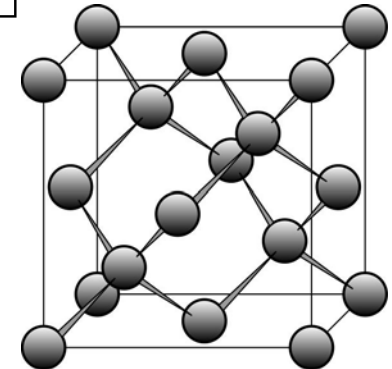
$$F_{hkl}^{(2)} = f \cdot \left[1 + e^{\pi i(k+l)} + e^{\pi i(h+l)} + e^{\pi i(h+k)} \right] \cdot e^{\pi i(h+k+l)/2}$$

Structure Factors:

$$F_{hkl} = F_{hkl}^{(1)} + F_{hkl}^{(2)} = f \cdot \overbrace{\left[1 + e^{\pi i(k+l)} + e^{\pi i(h+l)} + e^{\pi i(h+k)} \right]}^{4 \text{ or } 0} \cdot \overbrace{\left[1 + e^{\pi i(h+k+l)/2} \right]}^{2, 1 \pm i \text{ or } 0}$$

for hkl all even/all odd:

$$F_{hkl} = \begin{cases} 8f, & h+k+l = 4N \\ 4(1 \pm i)f, & h+k+l = 2N+1 \\ 0, & h+k+l = 4N+2 \end{cases}$$



As for *fcc*, allowed reflections have h, k, l all even or all odd.

However, some of these reflections are *kinematically forbidden*.

Zincblende Structure

Two distinct, interpenetrating *fcc* lattices offset by $\frac{1}{4}[111]$

First lattice: $\mathbf{d}_1 = \mathbf{0} = 0 \cdot \mathbf{a}_1 + 0 \cdot \mathbf{a}_2 + 0 \cdot \mathbf{a}_3 = (0, 0, 0)$

$$F_{hkl}^{(1)} = f_A \cdot \left[1 + e^{\pi i(k+l)} + e^{\pi i(h+l)} + e^{\pi i(h+k)} \right]$$

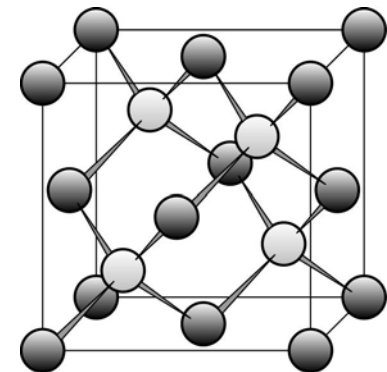
Second lattice: $\mathbf{d}_2 = 0.25a(\hat{\mathbf{x}} + \hat{\mathbf{y}} + \hat{\mathbf{z}}) = 0.25(\mathbf{a}_1 + \mathbf{a}_2 + \mathbf{a}_3) = (0.25, 0.25, 0.25)$

$$F_{hkl}^{(2)} = f_B \cdot \left[1 + e^{\pi i(k+l)} + e^{\pi i(h+l)} + e^{\pi i(h+k)} \right] \cdot e^{\pi i(h+k+l)/2}$$

$$F_{hkl} = F_{hkl}^{(1)} + F_{hkl}^{(2)}$$

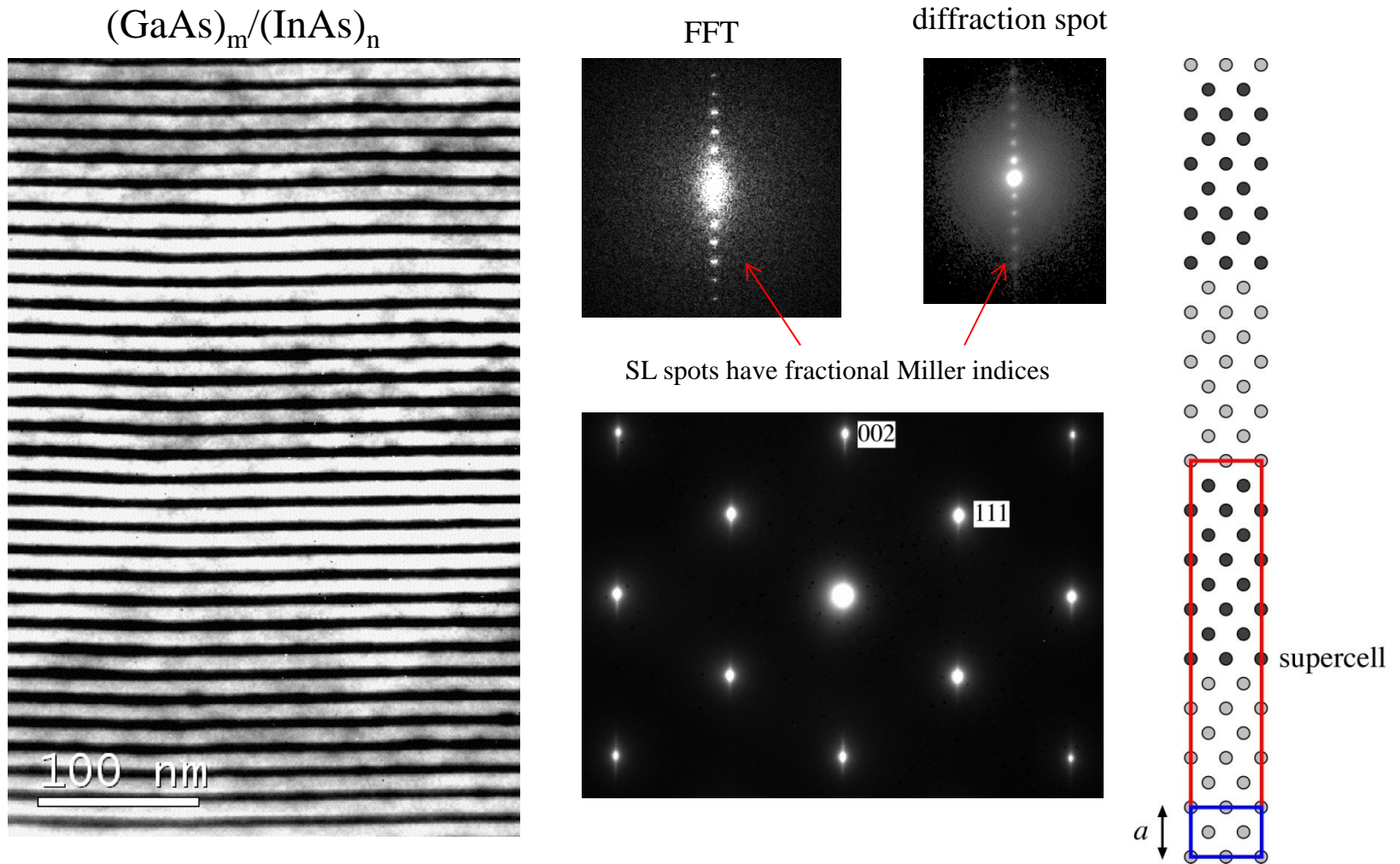
So,

$$F_{hkl} = \begin{cases} 4(f_A + f_B), & h+k+l = 4N \\ 4(f_A \pm if_B), & h+k+l = 2N+1 \\ 4(f_A - f_B), & h+k+l = 4N+2 \end{cases}$$



Allowed reflections have h, k, l all even or all odd.

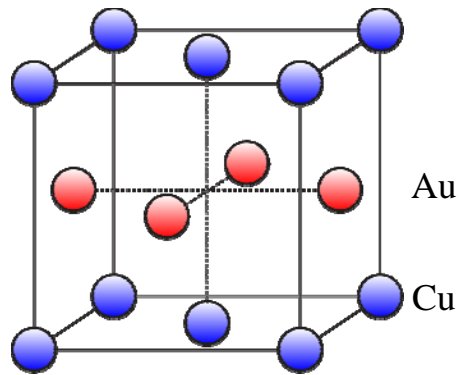
Superlattices



Large unit-cell built by ordering of a smaller unit cell

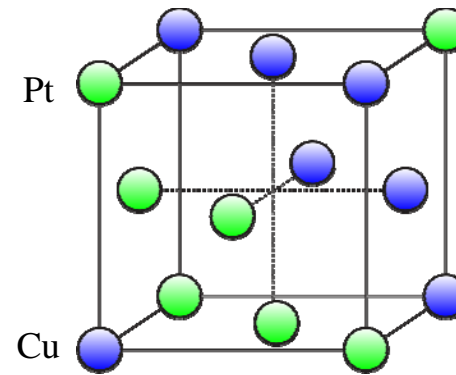
Atomic ordering in metal alloys

CuAu-I

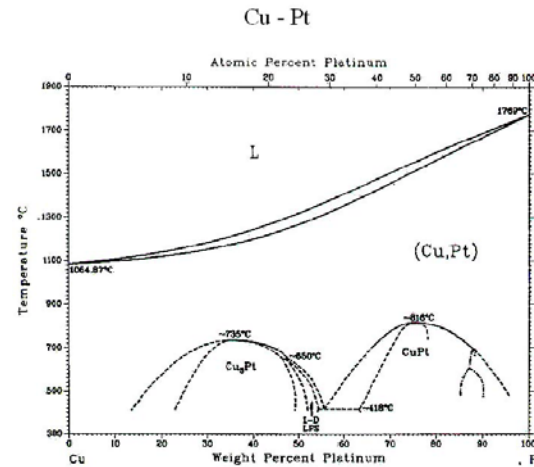
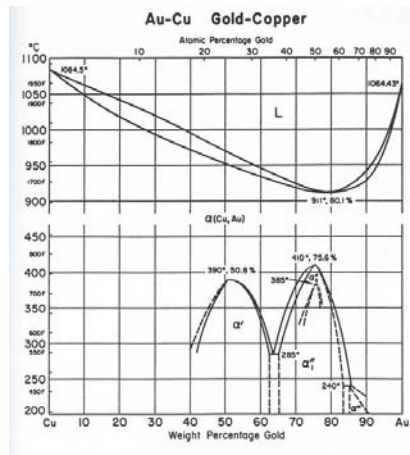


(001) superlattice

CuPt

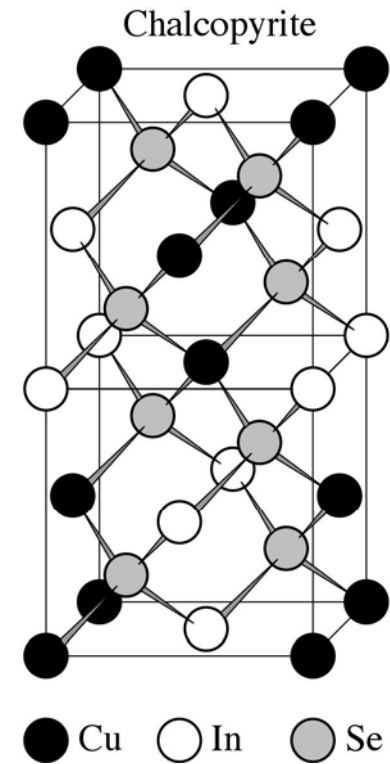
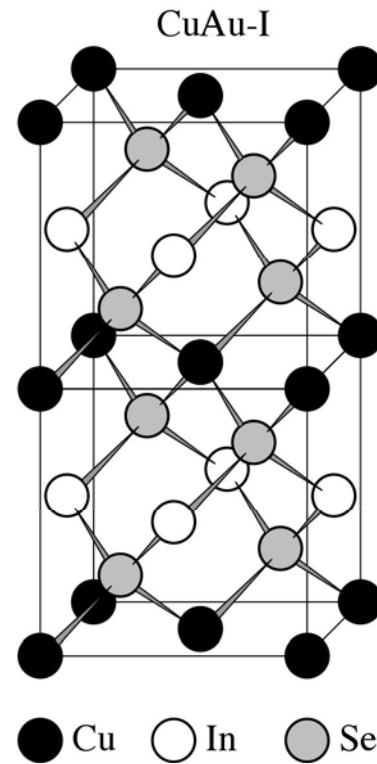
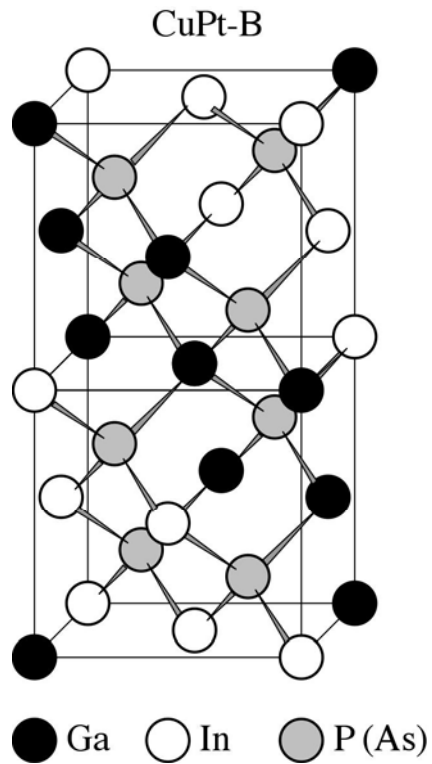


$\frac{1}{2}(111)$ superlattice



Specific atomic arrangements in normally random alloys

Atomic ordering in semiconductors



Sublattice ordering

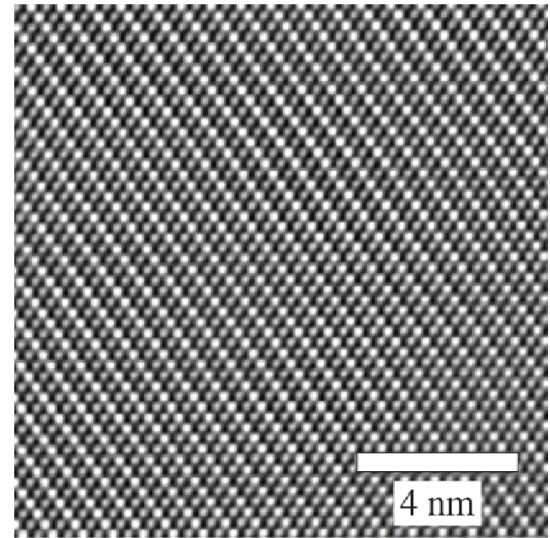
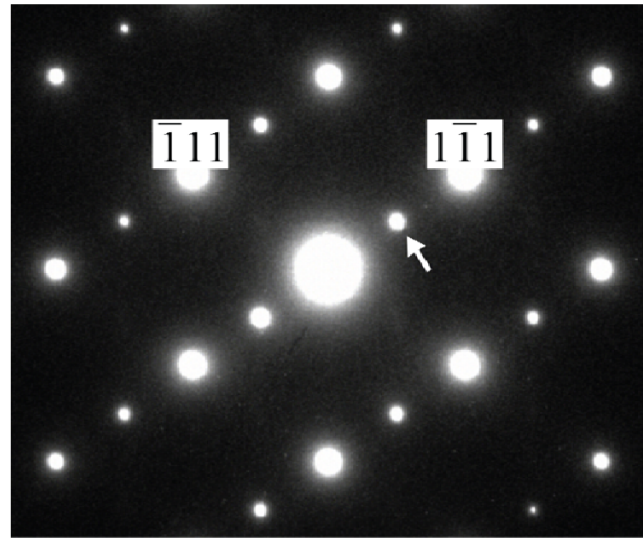
CuPt-Ordered GaInP₂

DP

HR

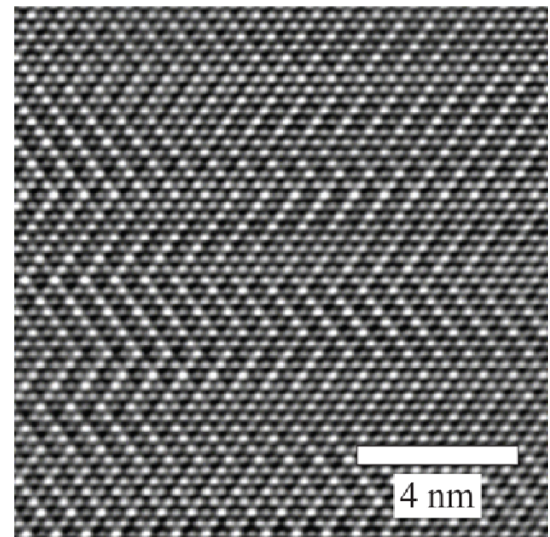
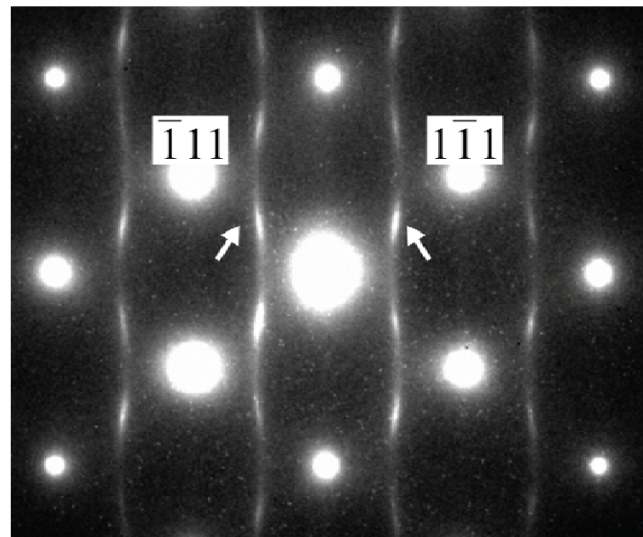
single-variant

[110]

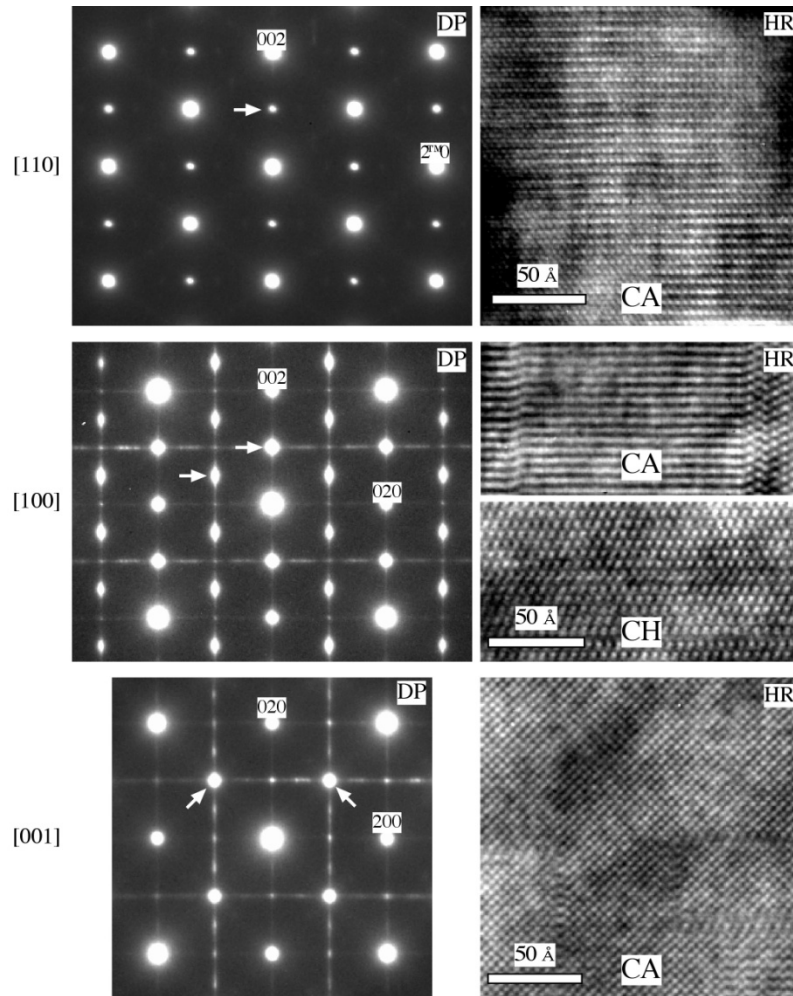


double-variant

[110]



Ordering in CuInSe_2



CuAu-I (CA) and
Chalcopyrite (CH)
Ordering

Example: Hexagonal close-packed structure

Basis vectors:

$$\mathbf{a}_1 = a\hat{\mathbf{x}}, \quad \mathbf{a}_2 = -\frac{a}{2}\hat{\mathbf{x}} + \frac{\sqrt{3}}{2}a\hat{\mathbf{y}}, \quad \mathbf{c} = c\hat{\mathbf{z}}$$

a is the nearest neighbor distance.

$c/2$ is the interlayer spacing.

The unit-cell volume is:

$$v = \frac{\sqrt{3}}{2}a^2c$$

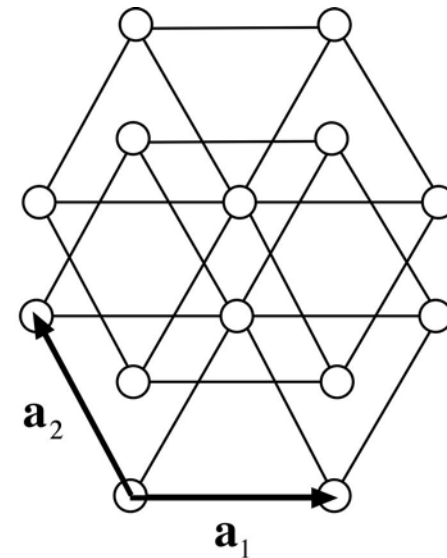
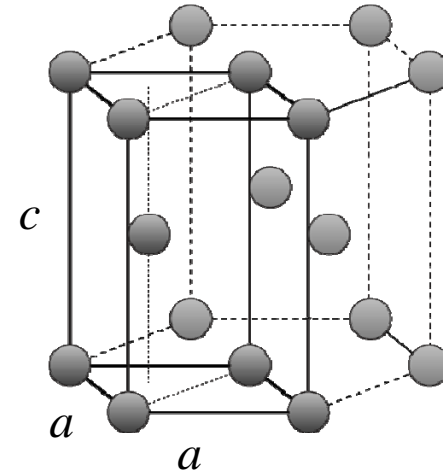
Two atoms in the unit cell:

First atom: f $\mathbf{d}_1 = (0, 0, 0)$

Second atom: f $\mathbf{d}_2 = \left(\frac{1}{3}, \frac{2}{3}, \frac{1}{2}\right)$

Structure Factors: $F_{hkl} = f \left[1 + e^{2\pi i \left(\frac{h}{3} + \frac{2k}{3} + \frac{\ell}{2} \right)} \right]$

Notice: $F_{00\ell} = 0$ for $\ell = \text{odd}$



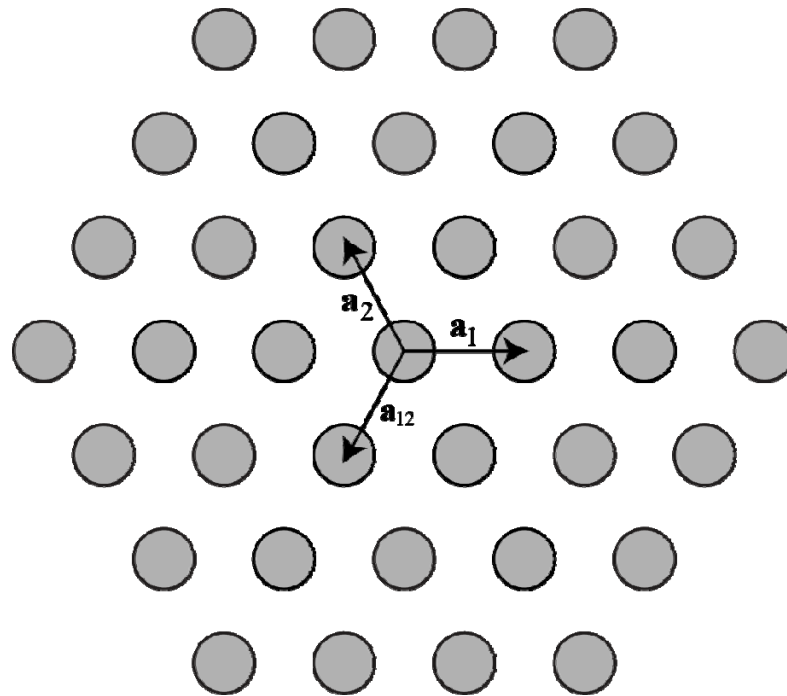
Indexing hexagonal crystals (I)

A direct-lattice vector could be written:

$$\mathbf{r}_{UVW} = U \cdot \mathbf{a}_1 + V \cdot \mathbf{a}_2 + W \cdot \mathbf{a}_3$$

There is a third, in-plane, symmetry equivalent vector:

$$\mathbf{a}_{12} = -(\mathbf{a}_1 + \mathbf{a}_2)$$



Indexing hexagonal crystals (II)

We could rewrite a vector as:

$$\begin{aligned} \mathbf{r}_{uvtw} &= u \cdot \mathbf{a}_1 + v \cdot \mathbf{a}_2 + t \cdot \mathbf{a}_{12} + w \cdot \mathbf{a}_3 = \mathbf{r}_{UVW} \\ &= u \cdot \mathbf{a}_1 + v \cdot \mathbf{a}_2 - t \cdot (\mathbf{a}_1 + \mathbf{a}_2) + w \cdot \mathbf{a}_3 \end{aligned}$$

$$\mathbf{r}_{uvtw} = (u - t) \cdot \mathbf{a}_1 + (v - t) \cdot \mathbf{a}_2 + w \cdot \mathbf{a}_3$$

$$U = u - t$$

$$V = v - t$$

$$W = w$$

We can pick: $t = -(u + v)$ \longrightarrow

$$u = (2U - V)/3$$

$$v = (2V - U)/3$$

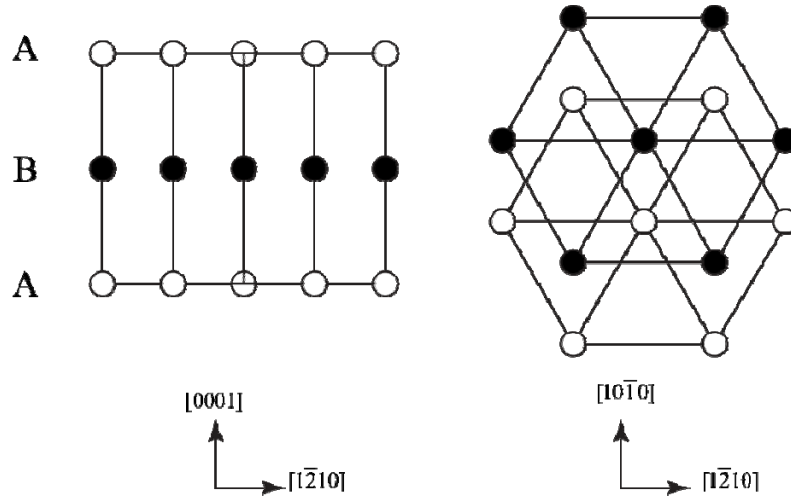
$$t = -(U + V)/3$$

$$w = W$$

$$\left. \begin{aligned} [100] &\rightarrow \frac{1}{3}[2\bar{1}\bar{1}0] \\ [010] &\rightarrow \frac{1}{3}[\bar{1}2\bar{1}0] \\ [\bar{1}\bar{1}0] &\rightarrow \frac{1}{3}[\bar{1}\bar{1}20] \end{aligned} \right\} \begin{array}{l} \text{symmetry} \\ \text{equivalent} \end{array}$$

Comparing close-packed structures: hcp and fcc

hcp

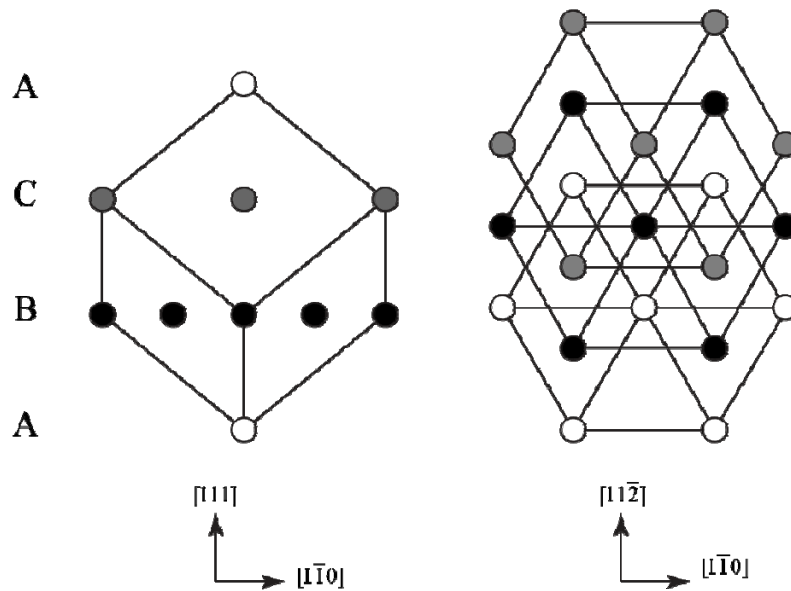


stacking sequence: AB/AB/AB/AB...

nearest-neighbor distance: $d_{n-n} = a^{(hcp)}$

interlayer spacing: $d_{002} = c/2$

fcc



stacking sequence: ABC/ABC/ABC...

nearest-neighbor distance: $d_{n-n} = a^{(fcc)} / \sqrt{2}$

interlayer spacing: $d_{111} = a^{(fcc)} / \sqrt{3} = \sqrt{\frac{2}{3}} d_{n-n}$

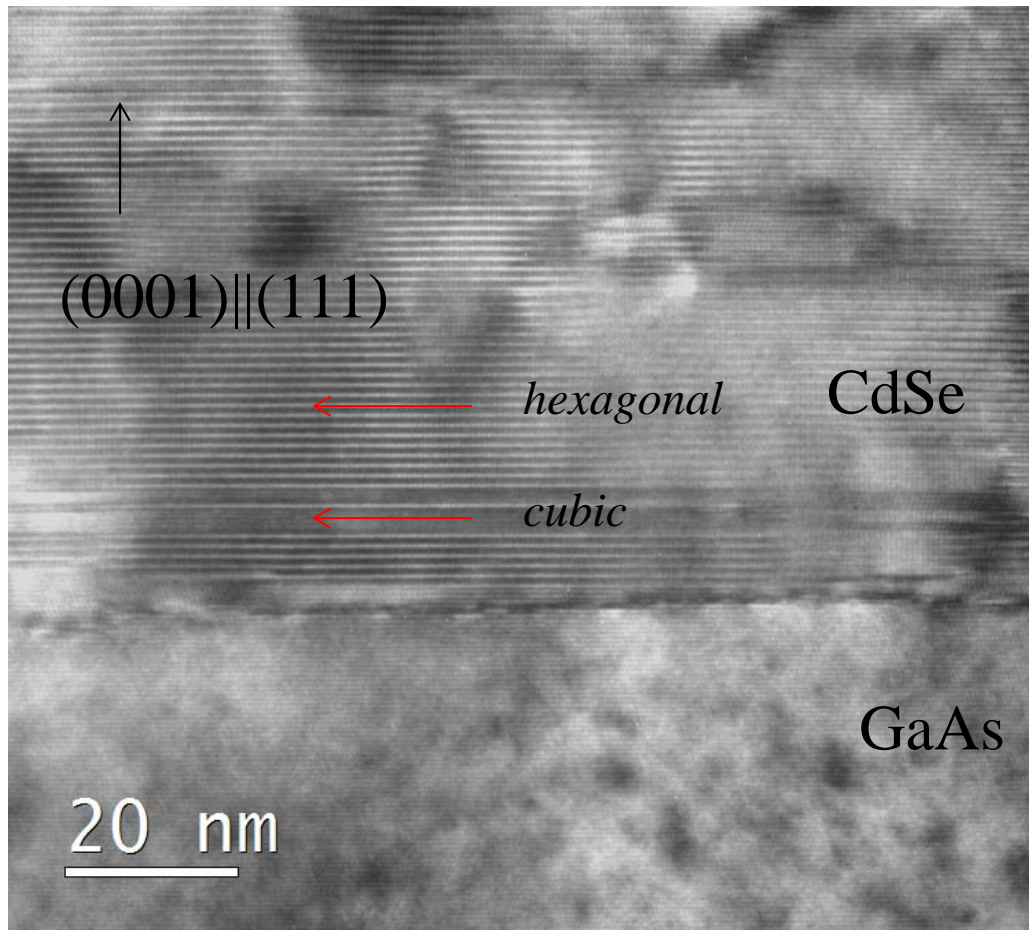
"Ideal" hexagonal close-packed

$$c^{(ideal)} = 2\sqrt{\frac{2}{3}}d_{n-n} = \sqrt{\frac{8}{3}}a^{(hcp)}$$

The two structures differ in stacking sequence.

Example: CdSe on (111) GaAs

HR



FFT

